



# Room temperature WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> sensors based on hierarchical microflowers for ppb-level H<sub>2</sub>S detection

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## ABSTRACT

Pristine Bi<sub>2</sub>WO<sub>6</sub> nanosheets and hierarchical WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers were prepared via a facile hydrothermal technique. WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers were assembled with thin nanosheets, where WO<sub>3</sub> nanoparticles were uniformly loaded. The H<sub>2</sub>S sensing properties of pristine Bi<sub>2</sub>WO<sub>6</sub> nanosheets and hierarchical WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers were systematically investigated at room temperature. 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers sensor displayed the best sensing property towards ppb-level H<sub>2</sub>S compared with all other samples. The enhancement was ascribed to the catalytic effect of WO<sub>3</sub> nanoparticles and the modified microstructure with large specific surface areas. Moreover, the n-n heterojunction structures also increase the thickness of electron depletion layer and potential barrier height, which could efficiently improve the sensing properties. The practical application of 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers sensor in detecting the volatiles of Pangasius was also studied in this work. Therefore, the hierarchical WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflower is a promising ppb-level H<sub>2</sub>S sensing material for environment monitoring and fish freshness detection.

## 1. Introduction

Environment and food safety have attracted more attention along with people's pursuit of healthy life [1,2]. Hydrogen sulfide (H<sub>2</sub>S) is colorless, malodorous and toxic [3–5]. It shows a great threat to human health, which will stimulate and damage respiratory organs and eyes even at a low concentration. The threshold limit value (TLV) of H<sub>2</sub>S for the industrial environment is defined as 10 ppm. Moreover, the acceptable level of the H<sub>2</sub>S for a human health condition should not exceed 100 ppb [6]. H<sub>2</sub>S is also a volatile biomarker gas resulting from the decomposition of sulfhydryl-containing amino acids in fish. The H<sub>2</sub>S trace amounts will reach or even exceed ppb levels when fish starts spoiling [7]. Additionally, the variation of H<sub>2</sub>S concentration (commonly in ppb level) in human exhaled breath can be utilized for diagnosing halitosis and real-time prevention of systemic diseases [8]. Hence, the development of high-performance gas sensing material for real-time and efficient ppb-level H<sub>2</sub>S detection is necessary for the environment, food safety and human health.

Metal oxides semiconductor materials have been comprehensively investigated as gas-sensitive materials, due to their high sensitivity, low cost and easy integration. There have been a series of metal oxide

nanomaterials for H<sub>2</sub>S detection. Deng et al. reported a microwave-assisted hydrothermal synthesized porous α-Fe<sub>2</sub>O<sub>3</sub> spheres based H<sub>2</sub>S sensor, showing a fast response to H<sub>2</sub>S with fine selectivity at 350 °C [9]. Peng et al. reported a CuO nanoparticles gas sensor, showing a response towards H<sub>2</sub>S at 150 °C [10]. Wang et al. obtained ZnO microspheres assembled with nanosheets, and the corresponding sensor exhibited responses to 5–100 ppm H<sub>2</sub>S at 70 °C [11]. Mokoena et al. prepared a non-stoichiometric NiO based H<sub>2</sub>S sensor, which displayed good sensing performance in dry air at 75 °C [12]. Although the single metal oxide shows good sensing properties, the high operating temperature and detection threshold limitation affect the performance stability and practical application. Several strategies have been explored to lower the operating temperature and to improve the gas molecules' adsorption capacity, for instance, constructing unique hierarchical nanostructure, doping low amount of elements, loading on low-dimensional materials and fabricating homo or hetero-junction structures [13–17]. The results revealed that mixed binary metal oxides-based gas sensing materials obtained a significant improvement in both physical and chemical properties. Hence, binary metal oxide semiconductors have become emerging gas sensing materials, such as ZnFe<sub>2</sub>O<sub>4</sub> [18] and BiVO<sub>4</sub> [19]. However, researchers seldom report room temperature bismuth

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tungstate ( $\text{Bi}_2\text{WO}_6$ ) gas sensors [20].

$\text{Bi}_2\text{WO}_6$ , as a typical Aurivillius-phase perovskite, possesses a sandwich structure composed of perovskite layers ( $\text{WO}_4$ )<sup>2-</sup> and ( $\text{Bi}_2\text{O}_2$ )<sup>2+</sup> layers. Additionally,  $\text{Bi}_2\text{WO}_6$  belongs to the n-type semiconductor with a bandgap of  $\sim 2.89$  eV, which has been successfully used as a photocatalyst due to its outstanding electron transfer properties and excellent photo absorption capacity [21,22]. However, there are few reports about  $\text{Bi}_2\text{WO}_6$  based gas sensors. Liu et al. synthesized a multilayer  $\text{Bi}_2\text{WO}_6$  micro-nano hierarchical structure via template-assisted hydrothermal method, and the sensor showed a response and high selectivity towards ppm level  $\text{H}_2\text{S}$  at 260 °C, which was ascribed to the existence of oxygen vacancies, the high specific surface areas and large pore volume [23]. The  $\text{Bi}_2\text{WO}_6$ -rGO gas sensor showed fine sensing performance to 250 ppb  $\text{H}_2\text{S}$  at 350 °C, which benefited from the p-n heterojunctions between  $\text{Bi}_2\text{WO}_6$  with well-dispersed rGO [24]. It is obvious that their operating temperatures were still high, as well as a little poor ppb-level  $\text{H}_2\text{S}$  sensing properties. Hence, it is urged to develop novel micro-nano structured  $\text{Bi}_2\text{WO}_6$  based sensing materials for low temperature and high-performance detection.

In this work, we used a facile hydrothermal method with sodium dodecylbenzene sulfonate (SDBS) as an anionic surfactant to prepare  $\text{Bi}_2\text{WO}_6$  hierarchical microflowers, and  $\text{WO}_3$  nanoparticles were decorated on  $\text{Bi}_2\text{WO}_6$  hierarchical microflowers. The characterization results showed  $\text{WO}_3$  nanoparticles can be loaded on the  $\text{Bi}_2\text{WO}_6$  hierarchical microflowers assembled with nanosheets without any post-treatment. The room temperature sensing properties towards ppb level  $\text{H}_2\text{S}$  were investigated. The observed improvement in sensing properties was ascribed to the hierarchical structure and n-n heterojunction structure of  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ .

## 2. Material and methods

### 2.1. Synthesis of $\text{Bi}_2\text{WO}_6$ nanosheets and hierarchical $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ microflowers

All reagents (Shanghai Aladdin Biochemical Technology Co., Ltd.) were purchased as analytical grade reagents without further purification.  $\text{Bi}_2\text{WO}_6$  nanosheets and hierarchical  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  microflowers decorated with different mass fractions of  $\text{WO}_3$  nanoparticles were synthesized.  $\text{Bi}_2\text{WO}_6$  nanosheets are prepared via a hydrothermal method starting from  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  and  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  as precursors. The principle of the decoration is to add an excess of  $\text{WO}_3$  precursor (In this case,  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ ) during the formation of the  $\text{Bi}_2\text{WO}_6$  nanosheets via the following procedure: 0.97 g  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ , certain amounts of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  (ranging from 0.33 g to 0.99 g), and 0.0348 g SDBS were mixed into 35 mL deionized water, subsequently transferred the obtained suspension to 50 mL Teflon-lined stainless autoclaves after magnetically stirred for 1 h. It was then maintained at 180 °C for 24 h and cooled down naturally. These products were centrifuged with absolute ethanol 5 times. Finally, a series of as-synthesized samples were obtained after a drying treatment at 80 °C for 6 h in an oven. The final products synthesized with  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  of 0.33 g, 0.495 g (10 wt%), 0.66 g (20 wt%), 0.824 g (30 wt%) and 0.99 g (40 wt%) are denoted as  $\text{Bi}_2\text{WO}_6$ , 10  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ , 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ , 30  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  and 40  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ .

### 2.2. Characterization

The phase structures of all powders were determined via X-ray diffraction (XRD) using  $\text{Cu-K}\alpha$  radiation with a scanning rate of  $5^\circ \cdot \text{min}^{-1}$  (D8 Advance Bruker). The morphological characteristics and elements distribution of  $\text{Bi}_2\text{WO}_6$  were inspected using field-emission scanning electron microscopy (FESEM, S4800II Hitachi) equipped with energy-dispersive X-ray spectroscopy. The further nanostructure of products was observed by transmission electron microscopy (TEM, JEM-2100). High-resolution TEM (HRTEM) information was obtained using field

emission transmission electron microscopy (Tecnai G2 F30 S-TWIN, FEI). The chemical states of the elements were also studied by X-ray photoelectron spectroscopy (XPS, Thermo Fisher Scientific ESCALAB 250Xi). The BET specific surface areas were acquired through  $\text{N}_2$  adsorption/ desorption at 77 K. The pore sizes of samples were calculated by Barret-Joyner-Halenda (BJH) method using the isotherms (Autosorb IQ3, Quantachrome Instruments).

### 2.3. Fabrication and measurement of gas sensors

As shown in Fig. S1, the electrode slice was prepared based on the alumina substrate (6\*30 mm). The platinum paste was printed by screen printing onto the alumina substrates. The alumina substrates were calcined at 350 °C and then 850 °C for 20 min successively after being dried for 30 min at 80 °C. The Pt layer was composed of a heating element and measuring electrodes. The as-synthesized powders ( $\sim 3$  mg) were firstly mixed with deionized water ( $\sim 1$  mL) to obtain a slurry via using mortar and pestle, which was then brushed uniformly on the upper side of alumina substrate equipped with Pt electrodes on the same side. The gas sensor was obtained after drying and treatment at 150 °C for 12 h. The gas sensor test was carried out by a four-channel gas sensing testing instrument (Wuhan Huachuang Ruike Technology Co., Ltd.), which can collect electrical resistance signals of the corresponding channel with a DC power supply of 12 V and 4 A. The gas sensing tests were performed at room temperature ( $20 \pm 2$  °C) and the relative humidity RH is around 30 %. The other details of the testing system are displayed in Fig. S1.

The responses of n-type  $\text{Bi}_2\text{WO}_6$  and  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  are defined as  $R_a/R_g$ , which is the ratio of the stable sensor resistance in air to the stable sensor resistance in target gas. The response/recovery time is taken from the time for achieving 90 % change of resistance values.

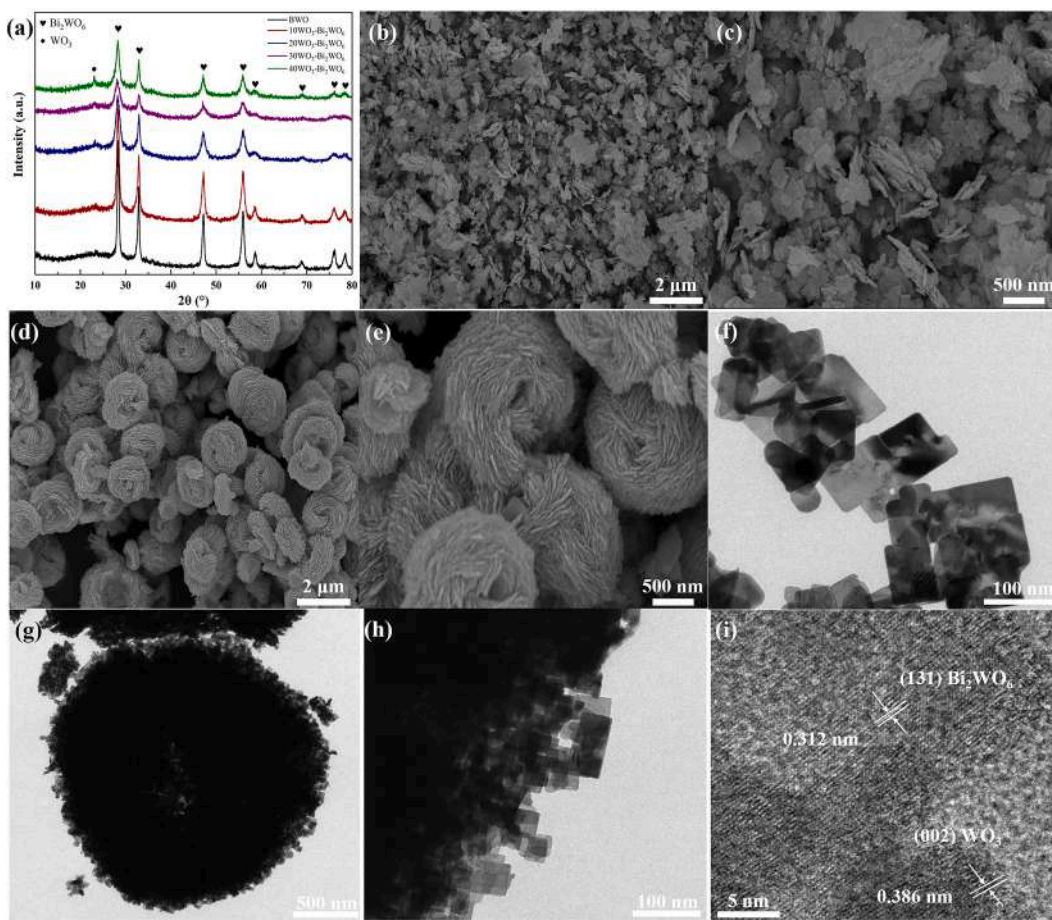
The fish freshness detection is measured on a homemade testing system. As shown in Fig. S2, 10 g fresh Pangasius fillet purchased from a local supermarket was firstly defrosted and then placed in a 1.25 L closed scrubber at ambient conditions and detected the released  $\text{H}_2\text{S}$  over different storage periods (0, 1, 2, 3, 4, 5, 12 and 24 h).

## 3. Results and discussion

### 3.1. Characterization results

XRD patterns of all samples from  $10^\circ$  to  $80^\circ$  are displayed in Fig. 1a. These distinct diffraction peaks observed at around  $28.31^\circ$ ,  $32.79^\circ$ ,  $47.16^\circ$ ,  $55.98^\circ$ ,  $58.61^\circ$ ,  $69.19^\circ$ ,  $76.20^\circ$ ,  $78.39^\circ$  and  $87.64^\circ$  are attributed to the standard orthorhombic-phase structure of  $\text{Bi}_2\text{WO}_6$  (JCPDS No. 39–0256). As the dopant amounts of  $\text{WO}_3$  nanoparticles increased, there exist obvious intensified monoclinic  $\text{WO}_3$  diffraction peaks at around  $23.08^\circ$  (JCPDS No. 43–1035), and the peak intensity of  $\text{Bi}_2\text{WO}_6$  decreased compared with pure  $\text{Bi}_2\text{WO}_6$ . The average crystal sizes of all samples are estimated using the Debye-Scherrer formula:  $D = 0.89\lambda/(\beta \cos \theta)$ , where  $\lambda = 0.15406$  nm,  $\beta$  is the full width at half maximum and  $\theta$  is the diffraction angle of the peaks. The crystal sizes of  $\text{Bi}_2\text{WO}_6$ , 10  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ , 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ , 30  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  and 40  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  are estimated to be 8.1, 11.6, 12.6, 8.5 and 8.7 nm, respectively. Besides, there are no extra obvious peaks, which implies the obtained  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  composites possess perfect phase structure and high purity.

The morphologies of all samples were observed through FESEM and TEM. In Fig. 1b and 1c, the FESEM images indicate that pure  $\text{Bi}_2\text{WO}_6$  products mainly consist of numerous nanosheets. In Fig. S3a and Fig. S3b, 10  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  showed microflowers with a diameter of around 2  $\mu\text{m}$  were assembled with thin nanosheets. The high magnification image demonstrated that there were amounts of  $\text{WO}_3$  nanoparticles loaded on the surface of microflowers. As the dopant amounts of  $\text{WO}_3$  nanoparticles continue to increase, it is interesting to observe that 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  sample showed the hierarchical microflowers decorated with nanoparticles in Fig. 1d and 1e. 30  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  and 40



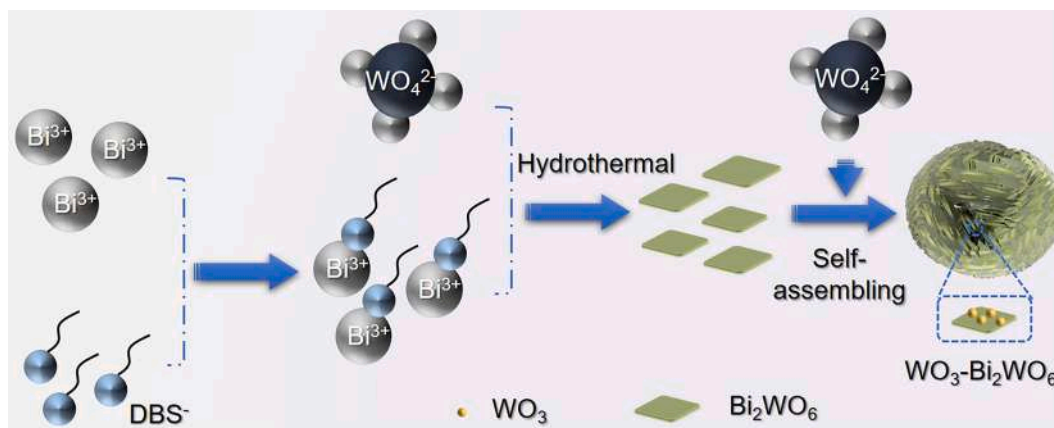
**Fig. 1.** (a) XRD patterns of pristine  $\text{Bi}_2\text{WO}_6$  and  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  composites; FESEM images of (b, c) pristine  $\text{Bi}_2\text{WO}_6$  and (d, e) 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ ; TEM images of (f) pristine  $\text{Bi}_2\text{WO}_6$ , (g, h) 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ , (i) HR-TEM image of 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ .

$\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  were displayed in Fig. S3(c–f). There is no difference in the average diameter of microflowers whereas the nanoparticles decorated on the surface became denser when the contents of  $\text{WO}_3$  exceeded 20 wt %, which may affect their gas sensing properties. Additionally, Fig. S4 showed the elemental mapping results of 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  microspheres, and all the elements are observed to be distributed uniformly.

As shown in Fig. 1(f and g), TEM analysis further confirms the microstructure of pure  $\text{Bi}_2\text{WO}_6$  and  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ . In Fig. 1f, pure  $\text{Bi}_2\text{WO}_6$  displays that thin nanosheets stacked on each other. Fig. 1g and 1 h reveal the nanosheets assembled spherical and hollow

microstructure of 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ , which can afford enough active sites for gas adsorption, as well as surface regions for gas-solid phase reaction [25]. In addition, the morphologies and sizes of hierarchical 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  microflowers are consistent with the observed FESEM images. As shown in Fig. 1i, the HRTEM image of 20  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$  further confirms the phase composition. The crystalline interplanar spacing of 0.312 nm and 0.386 nm are indexed to the (1 3 1) plane of  $\text{Bi}_2\text{WO}_6$  and (002) plane of  $\text{WO}_3$ , respectively.

According to the observation results of phase structure, morphology, and nanostructure, the synthesis mechanism of the hierarchical  $\text{WO}_3$ -



**Scheme 1.** Synthesis mechanism of pristine  $\text{Bi}_2\text{WO}_6$  and  $\text{WO}_3$ - $\text{Bi}_2\text{WO}_6$ .

$\text{Bi}_2\text{WO}_6$  microflowers is proposed in Scheme 1.  $\text{Bi}^{3+}$  cations and  $\text{DBS}^-$  anions are firstly bonded together to form Bi-DBS complexes resulting from the electrostatic interactions, which contributes to hindering the hydrolysis of  $\text{Bi}^{3+}$  and dispersion in the solution. With the introduction of  $\text{WO}_4^{2-}$ , it reacts with Bi-DBS to generate  $\text{Bi}_2\text{WO}_6$  nanosheets. Once the molar ratio of W to Bi element exceeds 1:2,  $\text{WO}_3$  nanoparticles will be generated and deposited on  $\text{Bi}_2\text{WO}_6$  nanosheets. Subsequently, they will self-assemble into  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  hierarchical microflowers for reducing the surface energy [26,27].

The element chemical states of pristine  $\text{Bi}_2\text{WO}_6$  and 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  were inspected by XPS. In Fig. 2a, the full survey XPS spectra of the two samples reveal that the four elements, Bi, W, O and C exist in both obtained products. Notably, C 1s peak was used as the reference for energy calibration in XPS measurements. Fig. 2b displayed two fitted peaks located at 163.95 and 158.63 eV in pure  $\text{Bi}_2\text{WO}_6$  spectra, corresponding to Bi 4f<sub>5/2</sub> and Bi 4f<sub>7/2</sub>, respectively. Meanwhile, there are also two similar characteristic peaks of 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$ , demonstrating the existence of  $\text{Bi}^{3+}$  in the products [28]. Fig. 2c displayed the high-resolution XPS spectrum of W 4f, which is fitted into two peaks of W 4f<sub>5/2</sub> at 36.98 eV and W 4f<sub>7/2</sub> at 34.83 eV, suggesting the presence of  $\text{W}^{6+}$  in the pristine sample [29]. As for 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$ , the characteristic spin-orbit peaks of Bi 4f and W 4f both displayed a slight shift to the high binding energy (Bi 4f: 0.23 and 0.26 eV; W 4f: 0.27 and 0.28 eV), which may be attributed to the influence of  $\text{WO}_3$  nanoparticles on the chemical environment of Bi and W [30]. Fig. 2d presents the O 1s spectra, which can be resolved into three peaks. The two peaks located at the binding energy of 529.51 eV and 530.53 eV correspond to the lattice oxygen, while the peak at 531.48 eV is attributed to the adsorbed oxygen species [31]. As for 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$ , there are three fitted peaks at the higher binding energy of 529.86, 530.55, and 531.78 eV. The shift to the higher binding energy of O 1s peaks (0.35, 0.02, and 0.30 eV) may be ascribed

to the reduction of the electron density around oxygen atoms, resulting from the formation of heterojunction in  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  composites [32,33]. Moreover, the intensity ratio of W-O to Bi-O peaks in  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  significantly increases compared with the pristine  $\text{Bi}_2\text{WO}_6$ , which also demonstrates the successful introduction of  $\text{WO}_3$  into the composite. All the above results confirm the existence of n-n heterojunction in hierarchical 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers.

The BET surface area of  $\text{Bi}_2\text{WO}_6$  and 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  samples were also obtained via the nitrogen adsorption-desorption isothermal technique. The isotherms plot of  $\text{Bi}_2\text{WO}_6$  nanosheets and 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers are displayed in Fig. 3. According to the isotherms and average pores diameters of  $\text{Bi}_2\text{WO}_6$  nanosheets and 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers, they all exhibit mesoporous structures [29]. The specific surface areas of  $\text{Bi}_2\text{WO}_6$  nanosheets and 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers were  $22.7 \text{ m}^2\cdot\text{g}^{-1}$  and  $47.5 \text{ m}^2\cdot\text{g}^{-1}$ , respectively. The results indicate that the microstructures and surface characteristics were modified after introducing  $\text{WO}_3$  nanoparticles. Moreover, the hierarchical structures with large surface area and porous structures are in favor of gas molecules' adsorption and transportation, which results in improving the sensing properties of 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$ . In addition, Figure S5 showed the  $\text{N}_2$  adsorption-desorption isothermal plots of 10, 30 and 40  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  composites. And as shown in Table S1, the specific surface areas of 10, 30 and 40  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  were 25.4, 52.4 and  $42.8 \text{ m}^2\cdot\text{g}^{-1}$ , respectively. The  $\text{WO}_3$  nanoparticles improve the specific surface areas, especially the  $\text{WO}_3$  amounts of 20, 30 and 40, which will provide more active sites for gas molecules adsorption and reaction. In Table S1, the average pore diameters of x wt%  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  (x = 0, 10, 20, 30 and 40) composites were 37.1 nm, 18.2 nm, 14.1 nm, 9.1 nm and 18.0 nm, respectively. And the corresponding total pore volumes are 0.192, 0.116, 0.168, 0.119 and  $0.193 \text{ cm}^3\cdot\text{g}^{-1}$ , respectively.

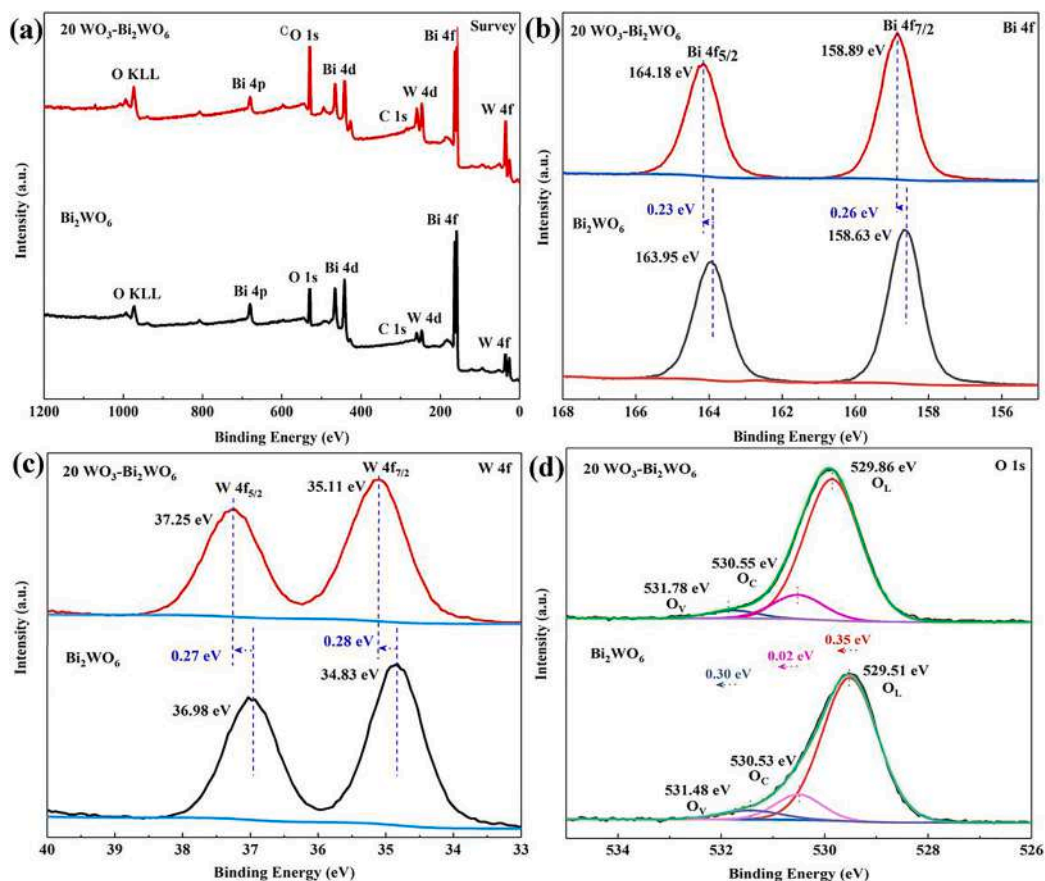


Fig. 2. Elements chemical states of pristine  $\text{Bi}_2\text{WO}_6$  and 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  composite determined by XPS: (a) survey, (b) Bi 4f, (c) W 4f and (d) O 1s.

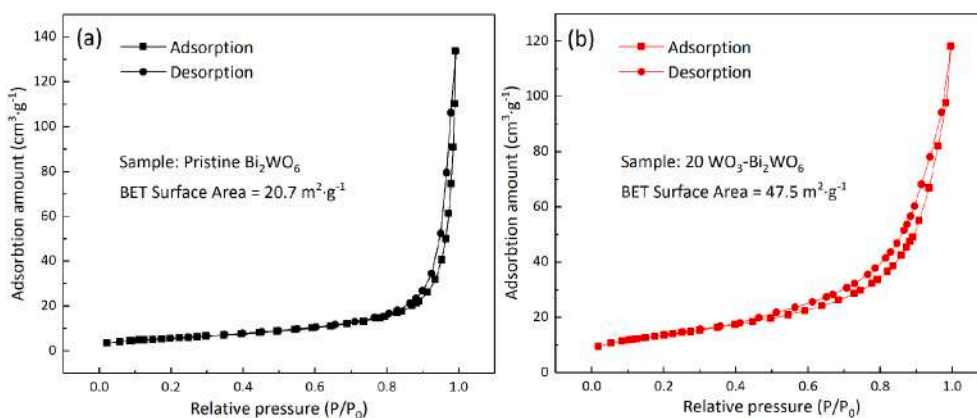


Fig. 3. Nitrogen adsorption–desorption measurement isotherm of (a) pristine  $\text{Bi}_2\text{WO}_6$  and (b)  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  composite.

### 3.2. Room temperature gas sensing performance

The sensing properties of  $\text{Bi}_2\text{WO}_6$  nanosheets and hierarchical  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers composites were evaluated at room temperature. The role of dopant amounts of  $\text{WO}_3$  nanoparticles on hierarchical  $\text{Bi}_2\text{WO}_6$  microflowers gas sensors was first investigated. Fig. 4a depicted that there is no response of pristine  $\text{Bi}_2\text{WO}_6$  nanosheets and  $10 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  towards  $50 \text{ ppb H}_2\text{S}$ , while  $20/30/40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  composites showed fine responses to ppb-level  $\text{H}_2\text{S}$ . Notably, the response of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  was 4.4, which was much higher than those of 30 and  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ . Furthermore, the response/recovery speed of three  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  was also measured at the same conditions. Among them,  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  exhibited the shortest response/recovery time ( $52 \text{ s}/119 \text{ s}$ ) to  $50 \text{ ppb H}_2\text{S}$  at room temperature. The introduction amounts of  $\text{WO}_3$  nanoparticles play a key role in the sensing properties of  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  composites. Hence,  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  was chosen as the best  $\text{H}_2\text{S}$  sensing material.

To further understand the dynamic response behavior of  $\text{WO}_3\text{-Bi}_2\text{WO}_6$ , Fig. 5a and 5b display the transient response/recovery curves of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ ,  $30 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  and  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  to  $2\text{--}50 \text{ ppb H}_2\text{S}$  at room temperature.  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  showed a much higher response than those of  $30 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  and  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ , revealing that the response of  $\text{Bi}_2\text{WO}_6$  based composite to  $\text{H}_2\text{S}$  was significantly improved via compounding with  $20 \text{ wt}\%$   $\text{WO}_3$  nanoparticles.  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  showed a large sensing capacity to ppb-level  $\text{H}_2\text{S}$ , and it can detect  $2 \text{ ppb H}_2\text{S}$  with a response of 1.13. In Fig. 5c, the fitting curves of two  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  samples fitted a linear relationship between response

values and  $\text{H}_2\text{S}$  concentration. Moreover, these linear fitting curves showed that the correlation coefficients  $R^2$  were in the range of  $\sim 0.974\text{--}0.998$ , and  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  present the highest correlation coefficient. Additionally, the theoretical limit of detection (LOD) was calculated from the experimental and fitting data by using the following equation [18]:

$$D_L = 3(RMS_{\text{noise}}/k)$$

in which  $RMS_{\text{noise}}$  is the gas sensor noise calculated via root-mean-square deviation processing on the stable baseline data taken exposure to air in Fig. 5f,  $k$  is the values of the slope of corresponding linear fitting curves. As shown in Table S2, the theoretical LOD of all three gas sensors are around  $2 \text{ ppb}$ ,  $8 \text{ ppb}$  and  $7 \text{ ppb}$ , respectively.

For monitoring the environment or freshness detection practically, selectivity as an important sensing characteristic should be evaluated. In Fig. 5d,  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  sensor was investigated using four interfering gases including  $\text{NH}_3$ ,  $\text{SO}_2$ ,  $\text{CO}$  and  $\text{H}_2$  of  $100 \text{ ppb}$  at the same condition. The response value was 7.86 towards  $\text{H}_2\text{S}$ , while the responses to  $\text{SO}_2$ ,  $\text{NH}_3$ ,  $\text{CO}$  and  $\text{H}_2$  were around 1.86, 1.11, 1.16 and 1.15, respectively. It can be concluded that the  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  microflowers sensor has an excellent selectivity, which is mainly attributed to two aspects. First, due to the lower bond dissociation energy of H-S in  $\text{H}_2\text{S}$  molecules compared to other target gases, it can be easily broken and interact with surface chemisorbed oxygen ions [34]. Second, the formation of n-n heterojunction at the interfaces of  $\text{WO}_3$  and  $\text{Bi}_2\text{WO}_6$  is also favorable for  $\text{H}_2\text{S}$  oxidation on the surface [35]. The humidity effect of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  microflowers sensor under a relative humidity of 30%, 50% and 70%

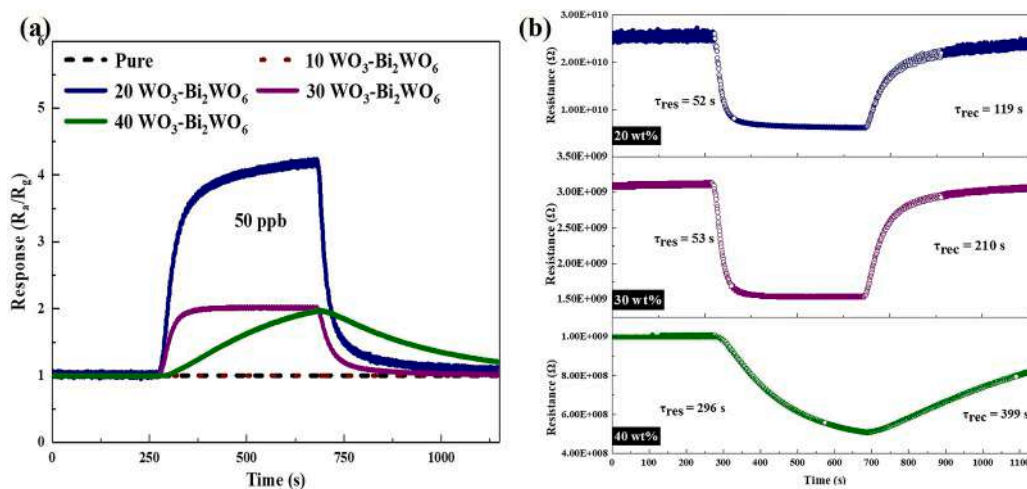
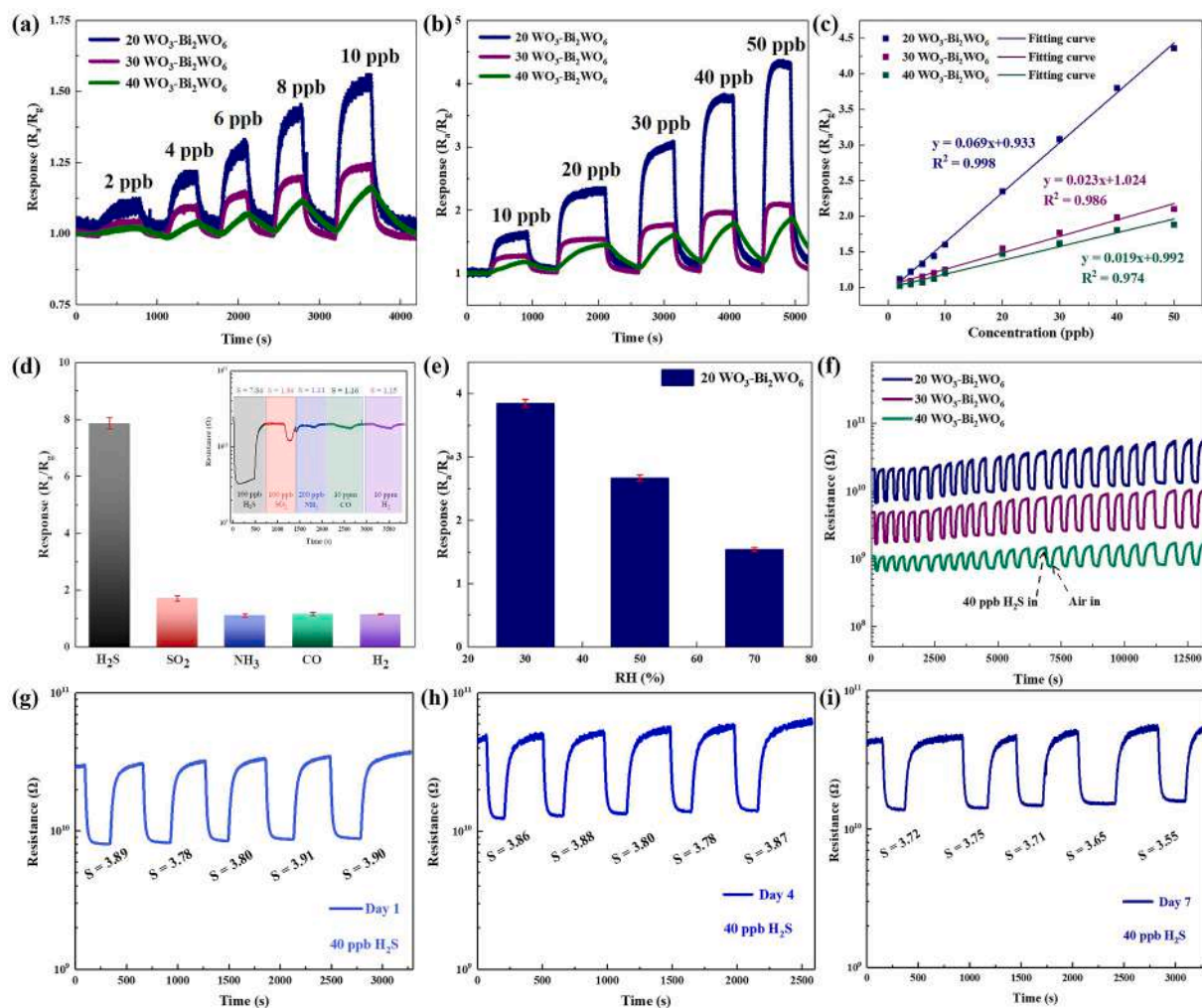


Fig. 4. (a) Responses to  $50 \text{ ppb H}_2\text{S}$  at room temperature; (b) dynamic response/recovery curves of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ ,  $30 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  and  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  to  $50 \text{ ppb H}_2\text{S}$  at room temperature.



**Fig. 5.** (a, b) Dynamic sensing performance of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ ,  $30 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  and  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  to 2–50 ppb  $\text{H}_2\text{S}$  at room temperature; (c) response values and the fitted curves of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ ,  $30 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  and  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  versus  $\text{H}_2\text{S}$  concentration at room temperature; (d) selectivity of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  to 100 ppb target gases at room temperature; (e) effect of relative humidity on  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  to 40 ppb  $\text{H}_2\text{S}$ ; (f) repeatable characteristics of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  in 24 consecutive responses; (g–i) repeatable characteristics of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  in day 1, day 4 and day 7.

was also investigated. As shown in Fig. 5e, the response to 40 ppb  $\text{H}_2\text{S}$  is 3.86, 2.68 and 1.54, respectively. The dynamic resistance variations in various relative humidity are shown in Fig. S6. The decrease of both baseline resistance and response may be ascribed to the competition between water and oxygen molecules on the surface, resulting in less reaction of adsorbed oxygen and  $\text{H}_2\text{S}$ . To identify the repeatability of each sensor during continuous response-recovery cycles, we first measured the three effective gas sensors ( $20$ ,  $30$ ,  $40 \text{ WO}_3\text{-Bi}_2\text{WO}_6$ ) for ppb-level  $\text{H}_2\text{S}$  detection via controlling the gas concentration manually. The results during 24 cycles are shown in Fig. 5f, which demonstrate that the three sensors attained repeatable response in the continuous measurement to 40 ppb  $\text{H}_2\text{S}$ , moreover, it can be found the resistance variation of gas sensors did not decrease along with the testing time, so these sensors present promising and effective response to  $\text{H}_2\text{S}$  in the 24 cycles (the total testing time was around 13000 s). Furthermore, as for the best gas sensor, we test the stability of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  gas sensor during one week to 40 ppb  $\text{H}_2\text{S}$  at room temperature and relative humidity of around 30%. As shown in Fig. 5g–i, the response of  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  towards 40 ppb  $\text{H}_2\text{S}$  in Day 1, 4 and 7 show good repeatability. The slight variation of response values to 40 ppb  $\text{H}_2\text{S}$  indicates that all the  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  composites gas sensors present a stable response at room temperature. Hence, the  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  composites gas sensors show expected performance in repeatability.

The comparison of hydrogen sulfide sensing properties between  $\text{H}_2\text{S}$

sensors in literature and  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  sensor in this work is summarized in Table 1. Comparing with other  $\text{Bi}_2\text{WO}_6$  based gas sensors,  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  composite gas sensor presents significantly good room temperature  $\text{H}_2\text{S}$  sensing performance including higher response and faster response/recovery. Furthermore, compared with other metal oxide gas sensors in the literature,  $20 \text{ WO}_3\text{-Bi}_2\text{WO}_6$  still exhibits the best  $\text{H}_2\text{S}$  sensing properties and the obvious advantage of low working temperature. Hence, it is a competitive room temperature  $\text{H}_2\text{S}$  sensor for practical application.

### 3.3. Gas sensing mechanism

As shown in Fig. 6, the response procedure, and the energy band diagram of  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  heterojunction after contact are used to explain the sensing mechanism of  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  gas sensors. The response process of  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  sensing material was explained via gas–solid phase reaction models [42,43]. As described in Fig. 6a and 6b, the surface adsorbed oxygen ( $\text{O}_{2(\text{ads})}$ ) will capture electrons from  $\text{WO}_3$  and  $\text{Bi}_2\text{WO}_6$  sensing materials to form  $\text{O}_2^-(\text{ads})$  ions at a low working temperature. When the material detects  $\text{H}_2\text{S}$  gas, oxygen anions ( $\text{O}_2^-(\text{ads})$ ) would oxidize  $\text{H}_2\text{S}$  molecules into water ( $\text{H}_2\text{O}$ ) and sulfide dioxide ( $\text{SO}_2$ ), as well as releasing electrons ( $e^-$ ) back to the conduction band (CB) of  $\text{WO}_3$  and  $\text{Bi}_2\text{WO}_6$ .

Table 1

H<sub>2</sub>S sensing properties of metal oxides semiconductor sensors.

| Materials   | Working temp. (°C) | Concentration | Response (R <sub>a</sub> /R <sub>g</sub> ) | τ <sub>res.</sub> /τ <sub>rec.</sub> | Refs.     |
|---|--------------------|---------------|--|--------------------------------------|-----------|
| Bi <sub>2</sub> WO <sub>6</sub> nanostructure                             | 260 °C             | 5 ppb         | 2.3  | –                                    | [23]      |
| Bi <sub>2</sub> WO <sub>6</sub> /rGO                                      | 350 °C             | 250 ppb       | 2.7  | ~1000 s/350 s                        | [24]      |
| Bi <sub>2</sub> MoO <sub>6</sub> microflowers                             | 170 °C             | 100 ppb       | 4.0  | ~80 s/500 s                          | [5]       |
| BiVO <sub>4</sub> porous film   | 75 °C              | 500 ppb       | 1.4  | ~150 s/40 s                          | [36]      |
| WO <sub>3</sub> /Bi <sub>2</sub> W <sub>2</sub> O <sub>9</sub> nanoflakes | 92 °C              | 500 ppb       | 2.7  | –                                    | [37]      |
| WO <sub>3</sub> -SnO <sub>2</sub> nanowires                               | 200 °C             | 500 ppb       | 3.8  | –                                    | [38]      |
| NiO/WO <sub>3</sub> nanoparticles   | 100 °C             | 50 ppb        | 4.95 ± 2.9                                 | –                                    | [39]      |
| WO <sub>3</sub> /CuO  | 80 °C              | 300 ppb       | 2.14                                       | ~300 s/-                             | [40]      |
| ZnFe <sub>2</sub> O <sub>4</sub> nanofibers                               | 350 °C             | 100 ppb       | 3.3  | ~160 s/210 s                         | [41]      |
| 20 WO <sub>3</sub> -Bi <sub>2</sub> WO <sub>6</sub>                       | RT                 | 50 ppb        | 4.4  | 52 s/119 s                           | This work |

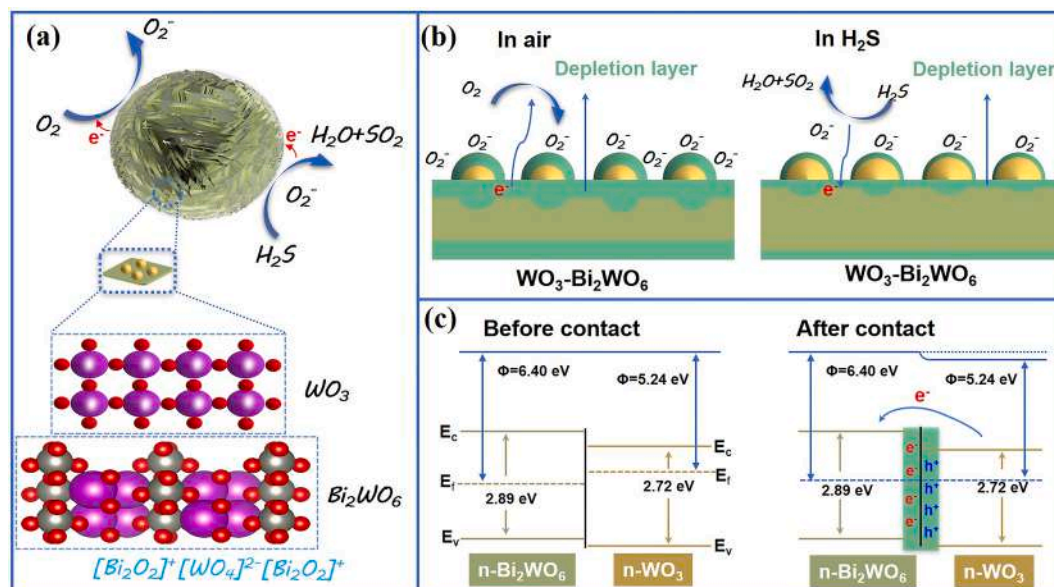
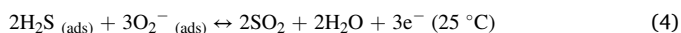


Fig. 6. (a, b) Schematic illustration of H<sub>2</sub>S sensing mechanism for hierarchical WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers and (c) Energy band diagram of 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> heterojunction structure.



The enhancement mechanism is also proposed. First, it is acknowledged that the sensing ability of semiconductor material is dependent on the surface oxygen adsorb capability [44]. As confirmed from the BET results, WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> showed a larger specific surface area, which could provide more effective reaction regions and contribute to enhancing the sensing efficiency. But when the WO<sub>3</sub> amounts exceed 20 wt%, according to the BET results, the worse sensing performance may be caused by the overloaded WO<sub>3</sub> nanoparticles, and the effect of specific surface area might not play the dominant role. Second, the enhanced gas-sensing performance of WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> is closely attributed to the hierarchical and hollow microflowers. The hollow structure behaves in hierarchical and porous characteristics, providing enough gas transportation channels for H<sub>2</sub>S molecules to diffuse into the WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> interior. Meanwhile, it also presents abundant active sites for the gas–solid phase reaction [45]. Thus, the sensing properties of WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> are improved. Third, it can be attributed to the difference in the energy level between n-type WO<sub>3</sub> nanoparticles (bandgap of ~2.72 eV) and n-type Bi<sub>2</sub>WO<sub>6</sub> (bandgap of ~2.89 eV), as shown in Fig. 6c. After the two materials are contacted, electrons would flow from WO<sub>3</sub> to

Bi<sub>2</sub>WO<sub>6</sub> until reaching an equilibrium state of the Fermi levels. Meanwhile, the electrons are easily captured to generate more chemisorbed oxygen ions and there will form a wider electron depletion layer at the n-n heterojunction interface and an increased potential barrier height. Once the WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> sensor is exposed to H<sub>2</sub>S, the oxidation reaction between chemisorbed oxygen ion and H<sub>2</sub>S molecules will occur rapidly, resulting in an enhanced response [46]. Fourth, WO<sub>3</sub> when added in a high amount is no more effective in creating n-n heterojunction because the nanoparticles are too close together and shield the underneath Bi<sub>2</sub>WO<sub>6</sub> phase.

#### 3.4. Practicability of the 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub>-based sensor

Monitoring the variation of H<sub>2</sub>S concentration can serve as a rapid and non-destructive method for fish freshness assessment. The fresh fish (pangasius) was selected as the target object and monitoring the electrical resistance change of 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> gas sensor during its spoilage process in 24 h at room temperature. Fig. 7a showed the correlation between sensor response values and various storage times at room temperature. The responses show a linear increase (R<sup>2</sup> = 0.9922) with the extension of dead fish storage time at 20 °C. Furthermore, the fabricated hierarchical 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> microflowers sensor also displayed fine transient response characteristics towards the volatiles of fish fillet. In Fig. 7b, when 20 WO<sub>3</sub>-Bi<sub>2</sub>WO<sub>6</sub> gas sensor detected the released gases from the fish stored for 0, 12 and 24 h at 20 °C, the response values were 2.6, 6.2 and 10.7, respectively. This trend further

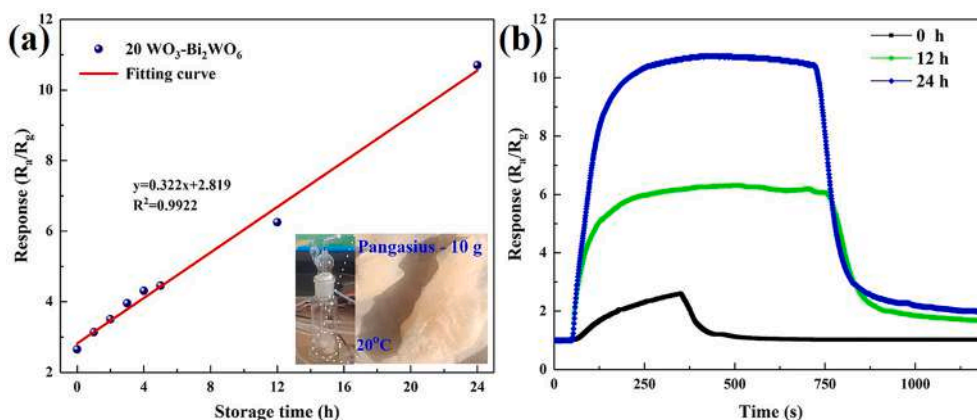


Fig. 7. (a) Responses of the 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  sensor as a function of storage time measured at 20 °C under exposure to odors from pangasius, (b) response values of 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  sensor to the volatiles from 10 g pangasius after storage for different times (0, 12, 24 h).

confirmed the obvious response changes of the sensor to different  $\text{H}_2\text{S}$  concentrations which were generated from the decomposition of sulfhydryl-containing amino acids during spoilage in pangasius [7].

#### 4. Conclusion

Pristine  $\text{Bi}_2\text{WO}_6$  nanosheets and hierarchical  $x$  wt%  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  ( $x = 10, 20, 30$  and  $40$ ) microflowers were prepared via a facile hydrothermal technique. The loaded  $\text{WO}_3$  nanoparticles modified the microstructure and morphology characteristics of  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers.  $\text{WO}_3$  nanoparticles were loaded on the surface of hierarchical  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers assembled with thin nanosheets. The pristine  $\text{Bi}_2\text{WO}_6$  and 10  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  showed no response to  $\text{H}_2\text{S}$  whereas 20/30/40  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  displayed improved sensing performance. Especially, 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers showed the best sensing properties, including a high response value of 4.4 to 50 ppb  $\text{H}_2\text{S}$ , while that of 30/40  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  were 2.0 and 1.9, respectively. The enhanced gas sensing properties may be ascribed to the catalytic effect of  $\text{WO}_3$  nanoparticles, the modified hierarchical structure with a large specific surface area and the  $n\text{-}n$  heterojunction structure. 20  $\text{WO}_3\text{-Bi}_2\text{WO}_6$  microflowers gas sensor realizes the monitoring of fish freshness, which presents a rapid and high-sensitivity analysis towards the released gases of pangasius. Therefore, this work proposed a facile synthesis process to fabricate room temperature ppb-level  $\text{H}_2\text{S}$  gas sensors for non-destructive evaluating fish freshness and real-time monitoring environment.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.cej.2021.132813>.

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